

Composite Electrolyte to Stabilize Metallic Lithium Anodes

Project ID: BAT273

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Overview – Composite Electrolytes to Stabilize Li Metal Anode

- Timeline
 - Start: October 2014
- Budget
 - \$400k FY18
 - \$400k FY19
- Technical barriers
 - Energy density (500-700 Wh/kg)
 - Cycle life, 3000 to 5000 deep discharge cycles
 - Safety
- Partners and collaborators
 - Oak Ridge National Laboratory (lead)
 - Collaborators:
 - ORNL collaborators – A. Westover, R. Sacci, F. Delnick
 - Ohara Corporation, CA
 - J. Schaefer, Univ Notre Dame
 - K. Hatzell, Vanderbilt Univ.
 - MERF for LLZO powders

Relevance and impact to VTO mission:

- Multi-year program plan identifies the Li metal anode and its poor cycling as the fundamental problem for very high energy Li batteries.
- Li metal confined and protected a solid electrolyte is the best route to safety & efficiency.
- Success of our composite electrolyte will enable:
 - Li-NMC and Li-S and Li-Air to meet the technical and cost objectives.

Objective for Li batteries → optimize energy density also, cycle life, processing cost and safety

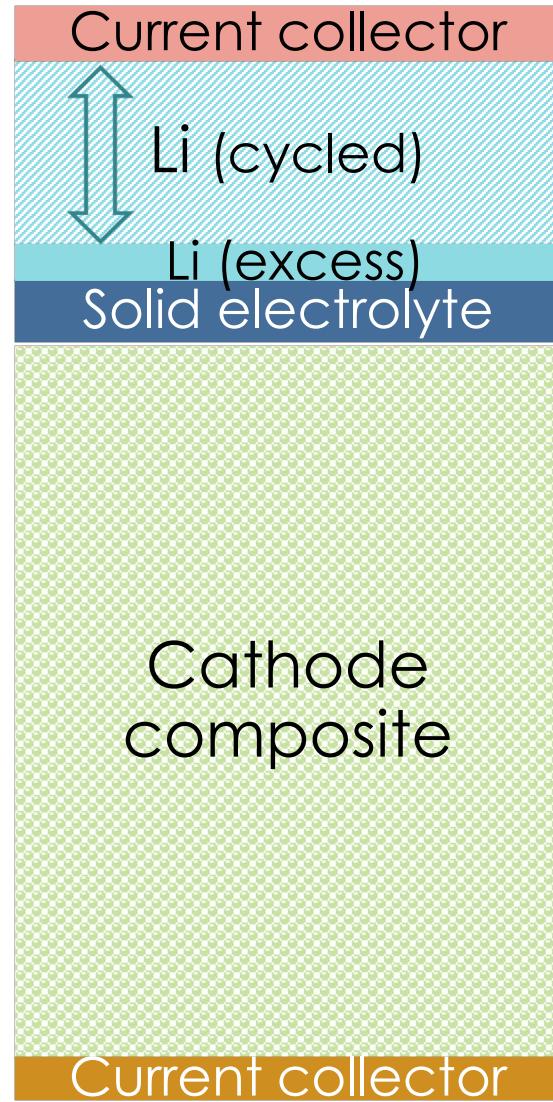
Thin or zero

**Equal capacity,
no excess**

**Thick for high
energy**

- Self supported
- Composites of active + SE

Thin film, not foil



Current collector - <10 µm

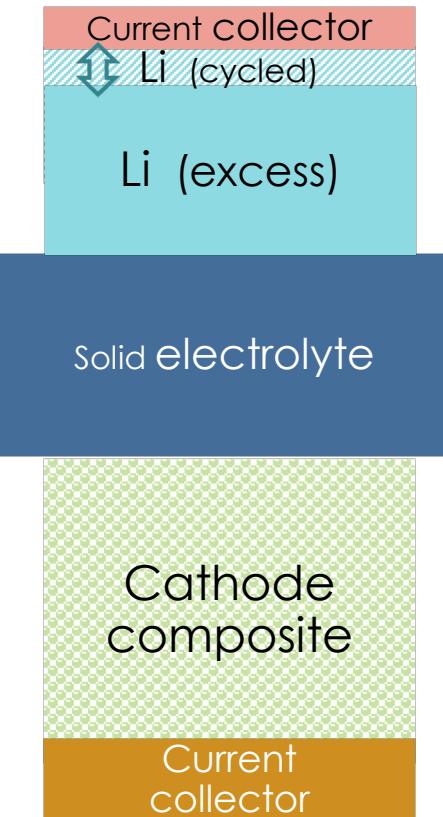
Li anode – 1-5 µm seed + 15-40 µm from cathode

Electrolyte separator– 1-20 µm

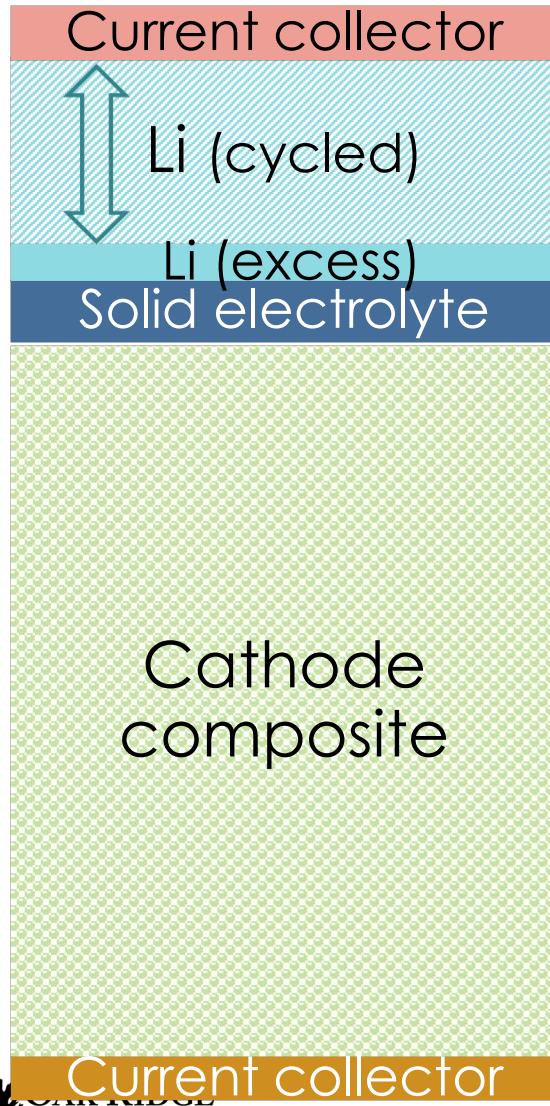
Cathode– 45-200 µm

Current collector – <10 µm

Starting point for most
Li metal batteries



Our approach: composite electrolytes → new materials
→ practical processing → interfaces → cells



Objectives – Study, determine

Composite solid electrolytes – thin, robust, conductive

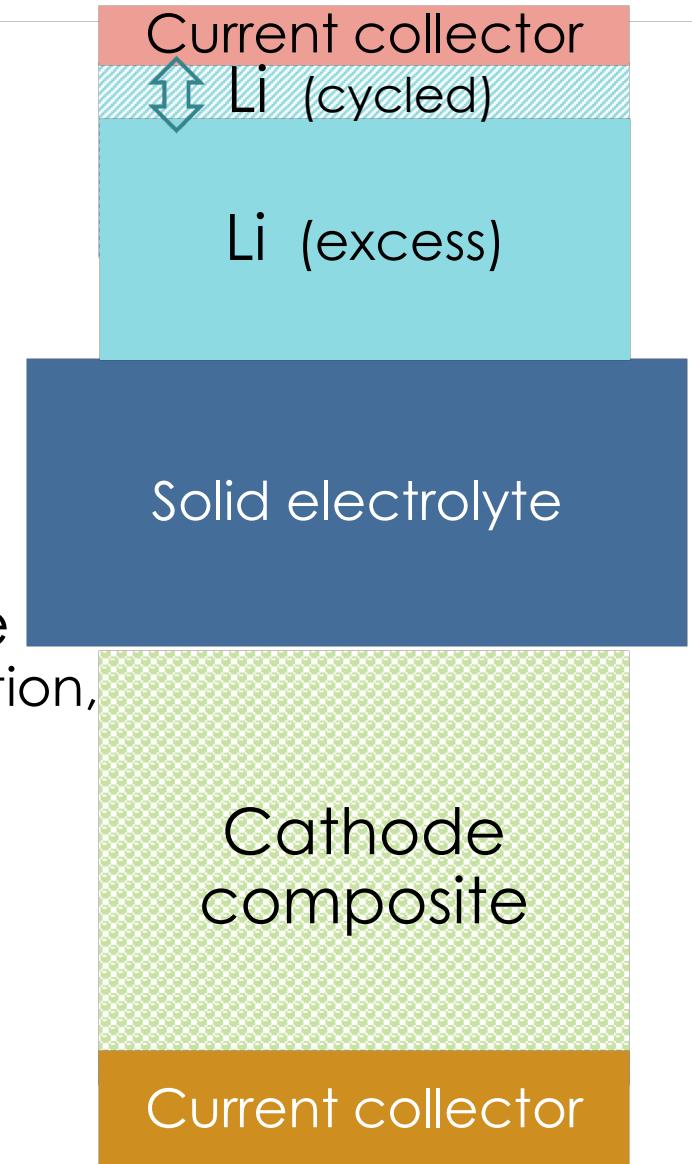
- Materials (polymer, ceramic)
- Structure
- Processing

Interfaces – stable, conductive

- Li – no reactions, no physical isolation, no dendrites
- HE/HV cathode

Full cells

- Cycle life
- Degradation processes



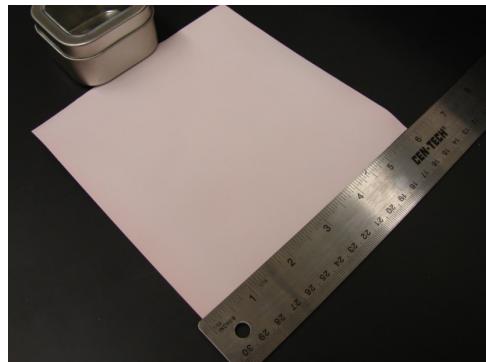
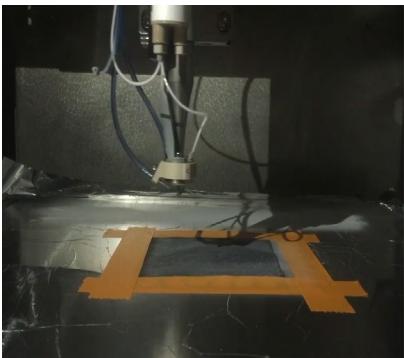
Starting point for most Li metal batteries

Milestones - FY19-20

Milestones: FY19-FY20	Target:	Status:
Fabricate gel composite electrolytes with a target room temperature conductivity of 1×10^{-4} S/cm. Thoroughly evaluate ion transport, thermal and structural properties.	Q3 FY19	
Expand composite materials portfolio to include non-PEO polymer electrolytes. Identify promising polymer ceramic systems with interfacial ASR less than 10 ohm .	Q4 FY19	Lowest ~50 Ω ASR to date
Develop methods to minimize interconnected composite electrolytes' interfacial resistance with lithium by varying polymer chemistry.	Q1 FY20	
Fabricate interconnected ceramic network with different ceramic chemistry and particle size to increase strength of the composite.	Q2 FY20	On track, with MERF LLZO
Investigate the trade-off between Li⁺ transference number and ionic conductivity of the gel composite electrolytes, and optimize it.	Q3 FY20	In progress
Fabricate full batteries using NMC cathode, composite electrolyte, and Li-metal anode. Identify cell failure mode .	Q4 FY20	In progress
Create chemical/physical bonding between polymer and interconnected ceramic network that leads to optimized interface to improve mechanical modulus and ionic conductivity.	Stretch FY20	

Composite Electrolytes to Stabilize Metallic Lithium Anodes: challenges and approaches

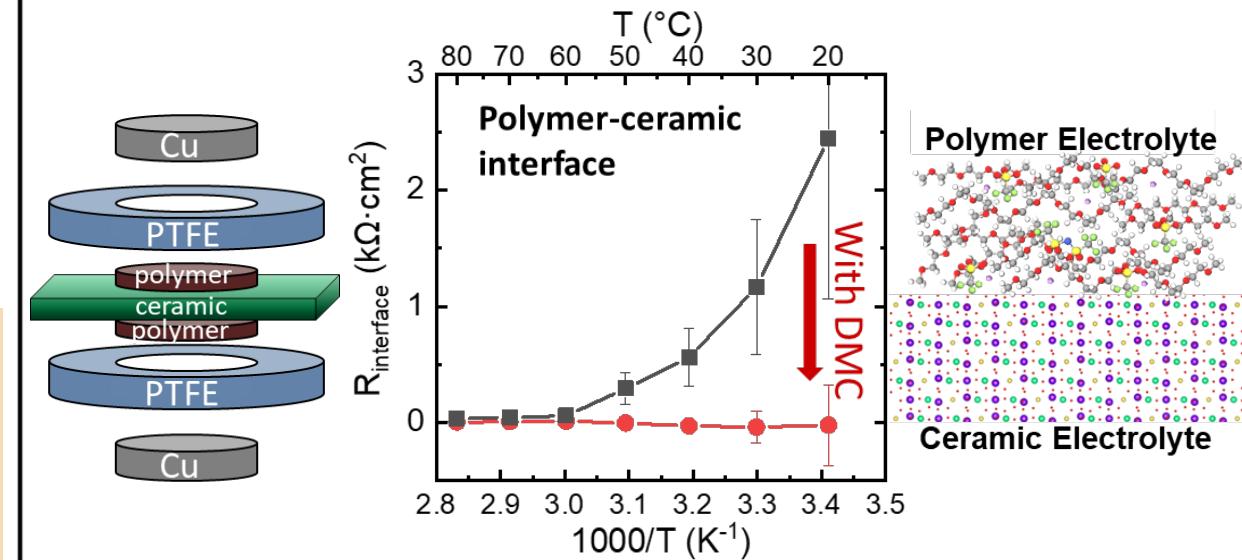
- Challenge 1:** fabricate composite with high ceramic loading → spray coating



Composites are:

- PEO + LiTFSI + Ohara LiCGC™ particles
- Aqueous slurry
- ~20-30 μm thick, after pressing
- Up to 50 vol% ceramic, dispersed particles
- Dry $\sim 10^{-8} \text{ S/cm}$, add TEGDME $\sim 10^{-5} \text{ S/cm}$

- Challenge 2:** quantify and minimize interfacial resistance between polymer and ceramic electrolytes → trilayer model

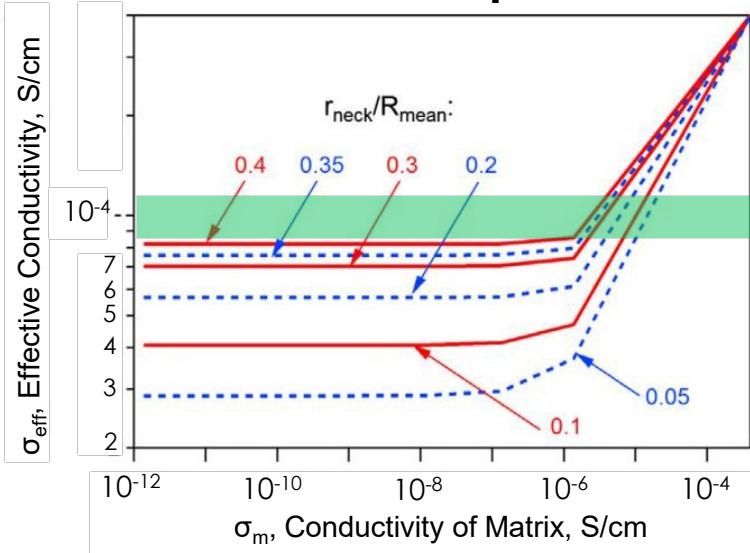


Are there other polymer electrolytes for better interfaces?

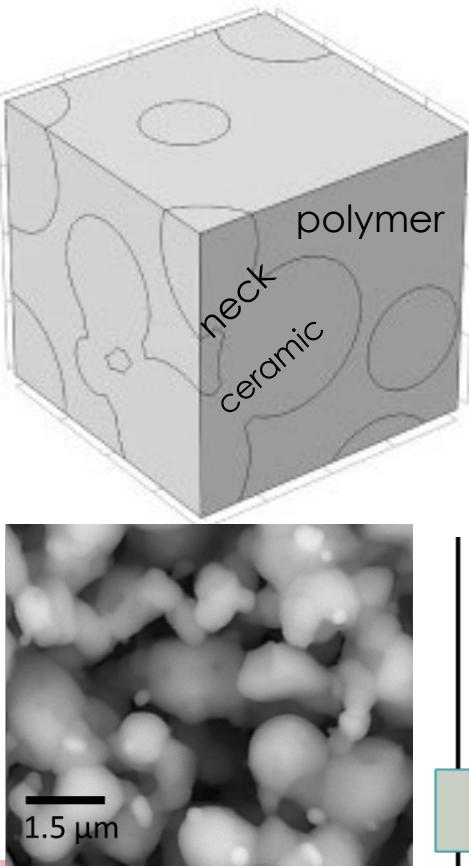
Composite Electrolytes to Stabilize Metallic Lithium Anodes: challenges and approaches

- **Challenge 3:** Fabricate composite to ensure connectivity of ceramic in composite

Critical geometry sintered neck radius/particle



Kalnaus, S., et al. *Journal of Power Sources* 2013, 241

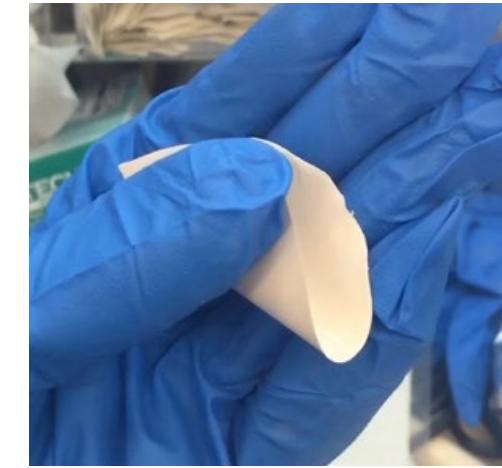
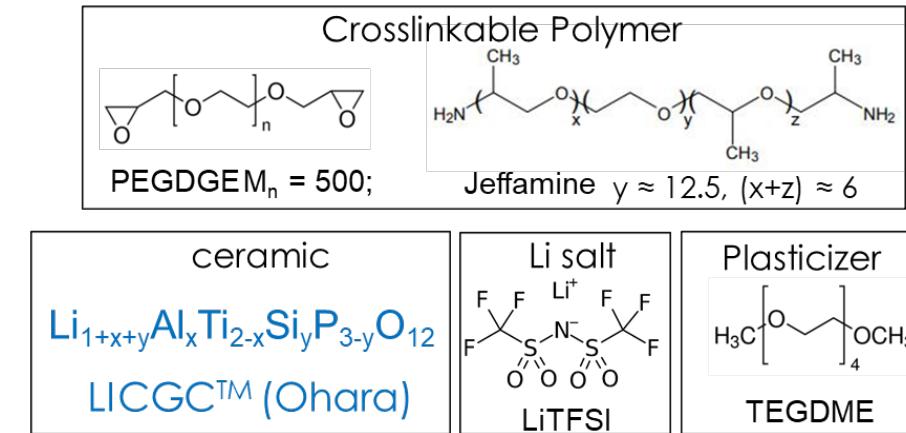
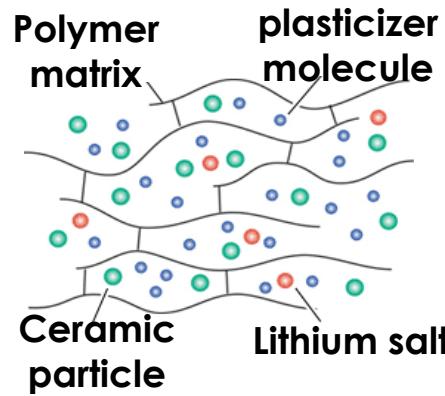


- **Challenge 4:** Integration to full cell with thin Li and dry cathode composite
 - Complex
 - Li anodes, thin films – thick foils
 - Multiple electrolyte materials, used in combination
 - Abandoned homemade NMC cathodes (temporarily) → good source of optimized dry polymer composite with LiFePO₄ cathode.

Improved polymers electrolytes

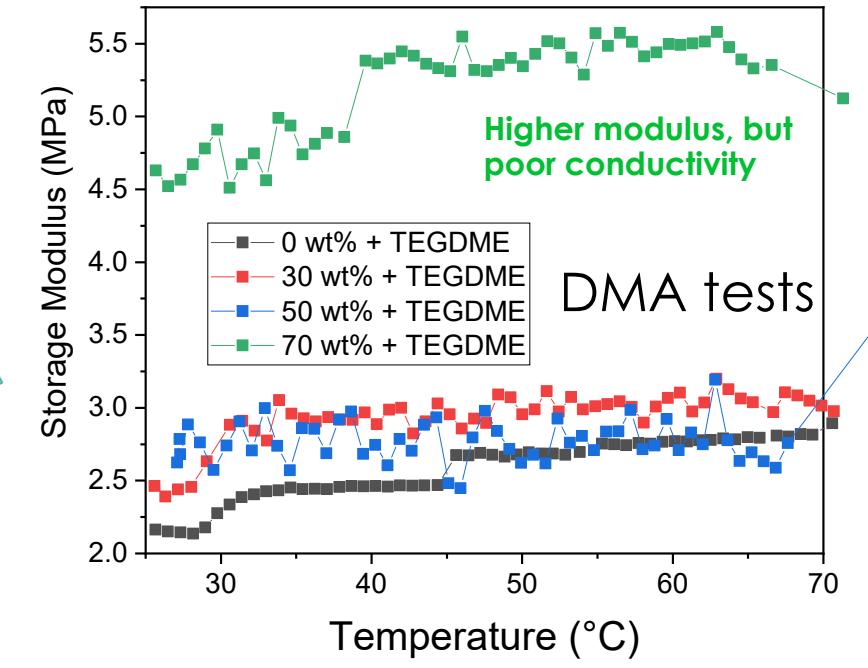
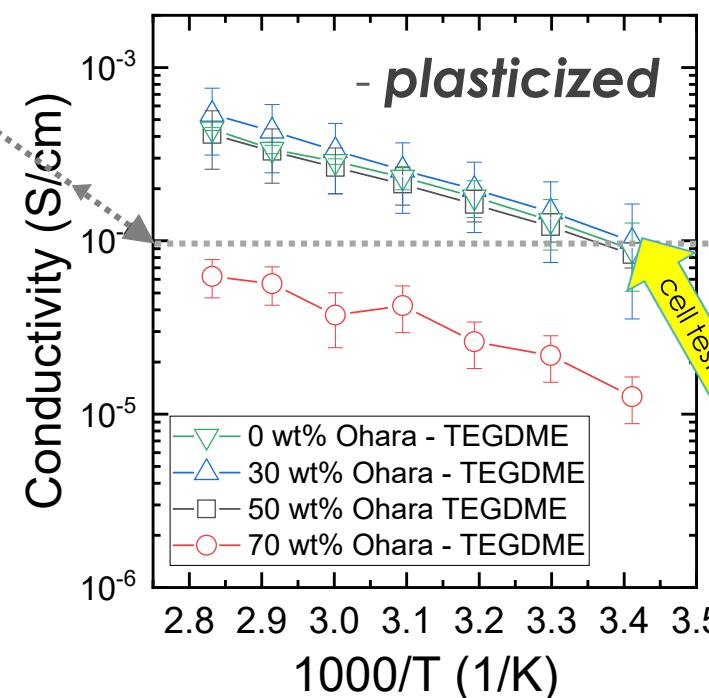
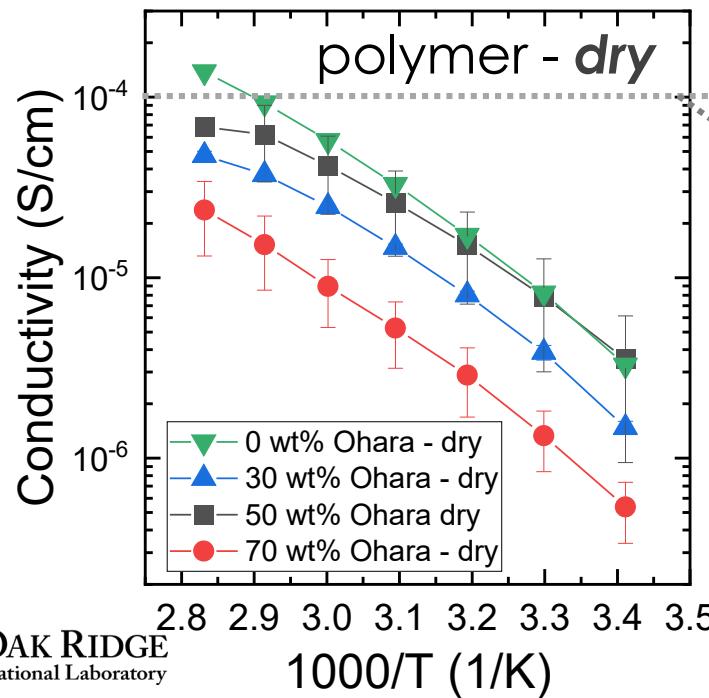
M. Palmer, et.al. N. Dudney, X. Chen, A Three-Dimensional Interconnected Polymer/Ceramic Composite of as a Thin Film Solid Electrolyte, *Energy Storage Materials*, 26, 242-249, 2020

Crosslinked Gel Composite – achieves both high Li⁺ ion transport and mechanical properties when plasticized



Good handling at 50wt%.

Brittle at 70wt%.

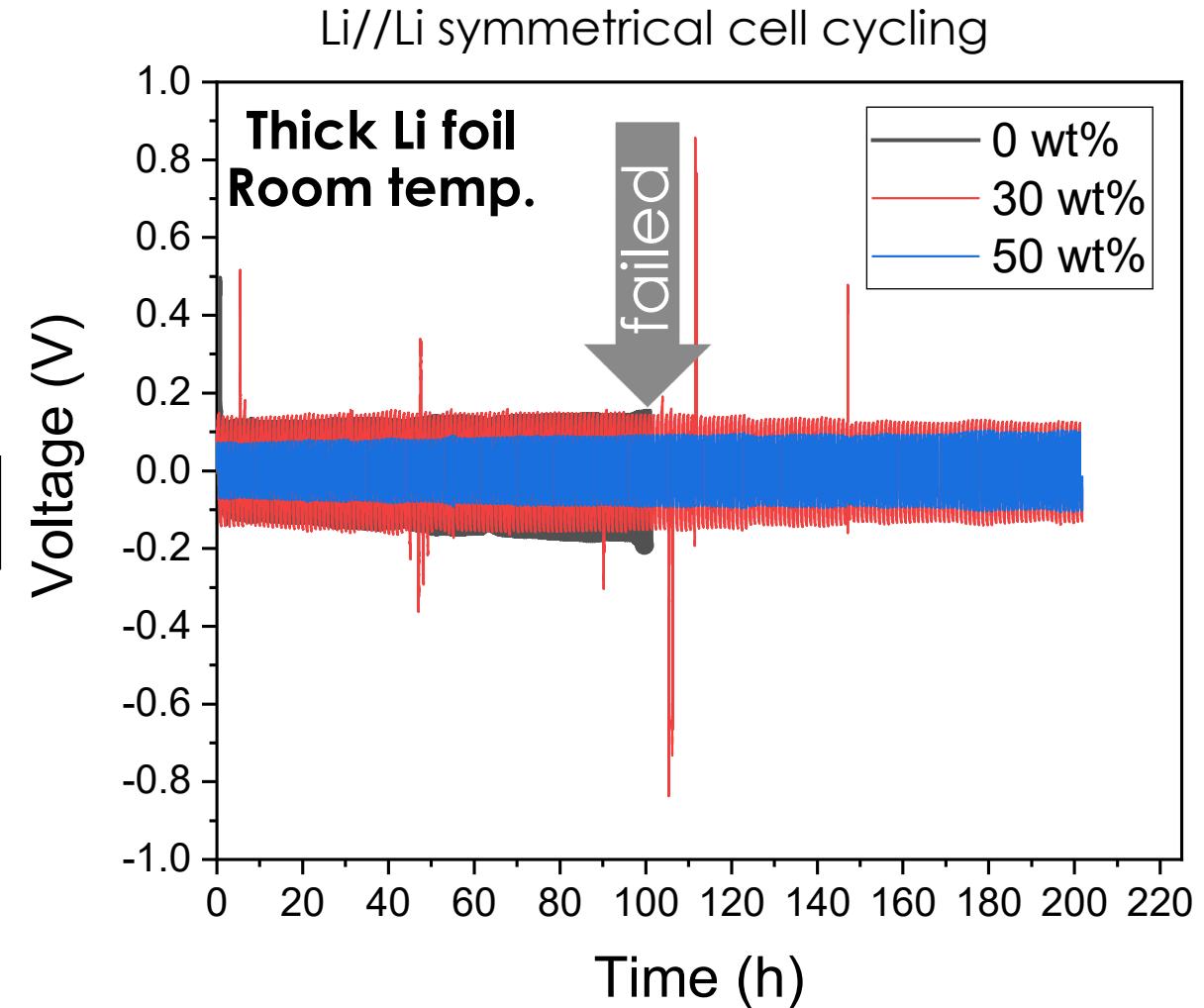
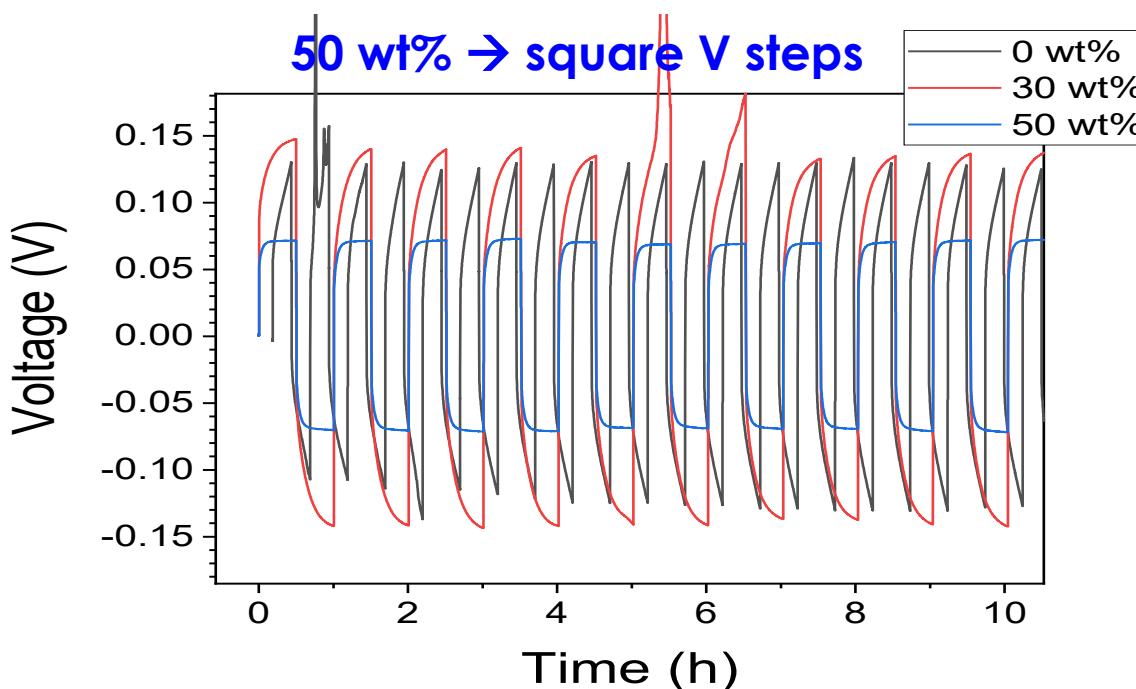


Crosslinked Gel Composite- Dispersed Ohara particles increases t_{Li^+} and cycling stability, with good conductivity

Accomplishment

Transference number measurement

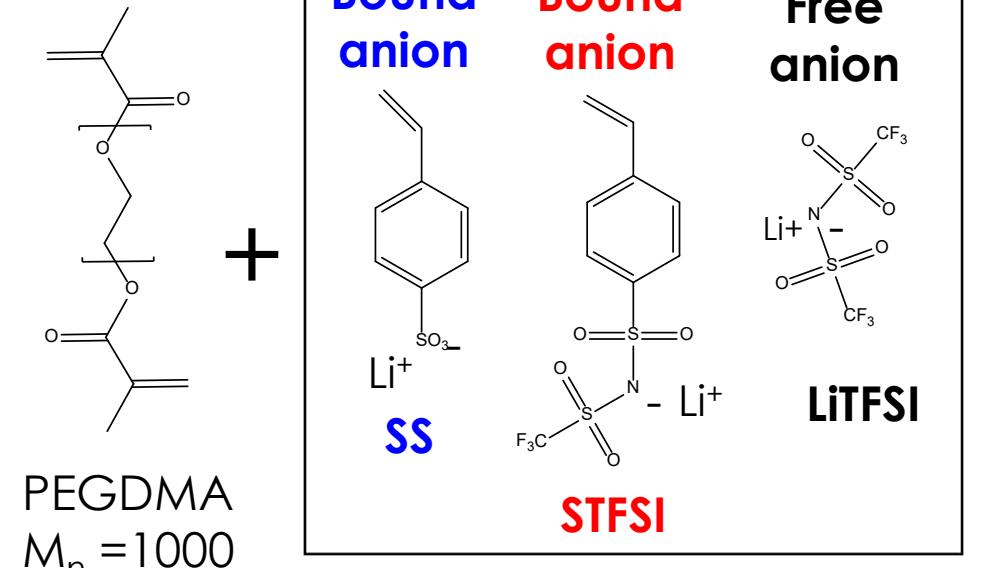
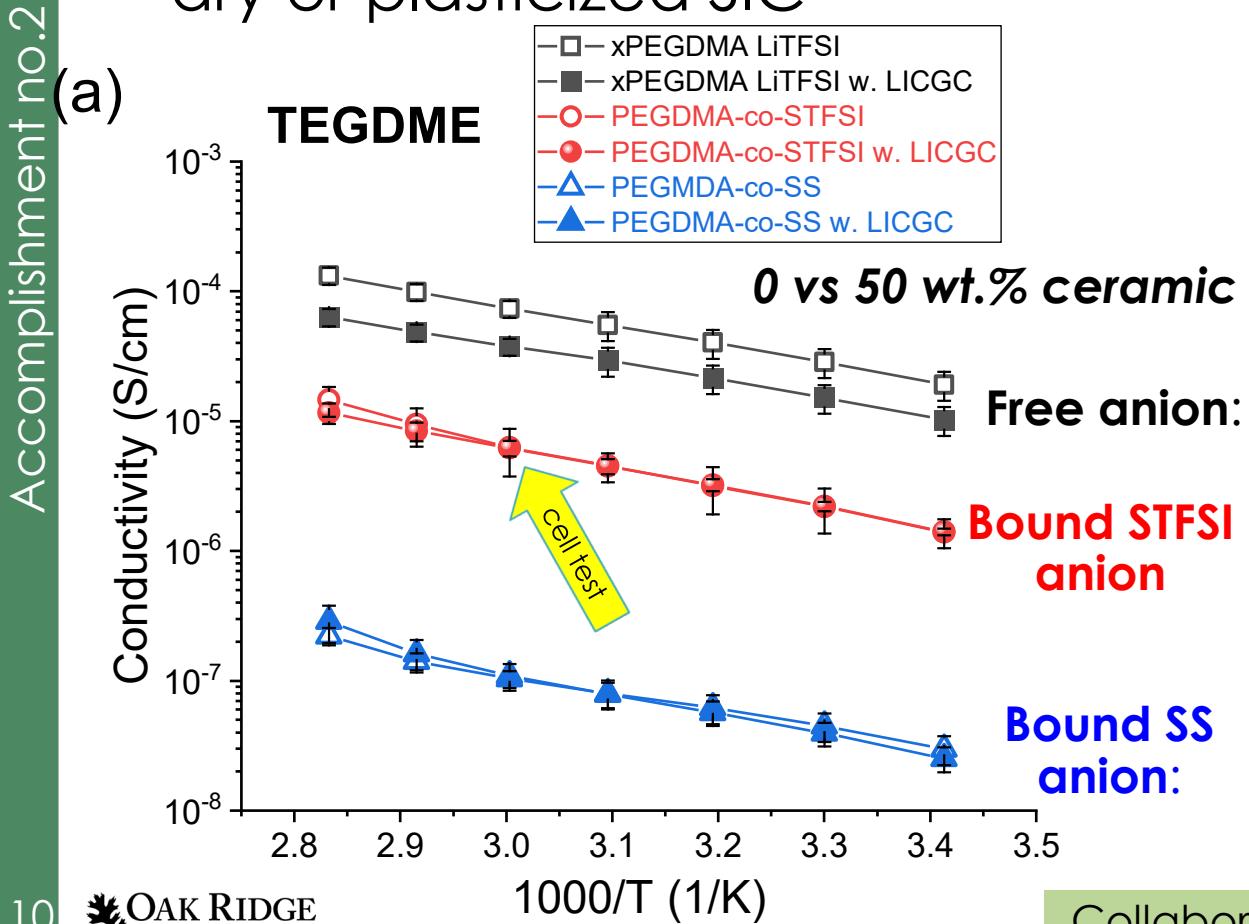
Samples	t_{Li^+}
Gel polymer	0.20
Gel with 30wt% LICGC™	0.16
Gel with 50 wt% LICGC™	0.44



Single-ion composite (SIC) – can have large ceramic loading, but plasticizer needed for high conductivity

- TEGME plasticizer (50-100X) increased conductivity
- 50wt.% ceramic does not impede conductivity of dry or plasticized SIC

But poor ionic conductors $\sim 1\mu\text{S}/\text{cm}$

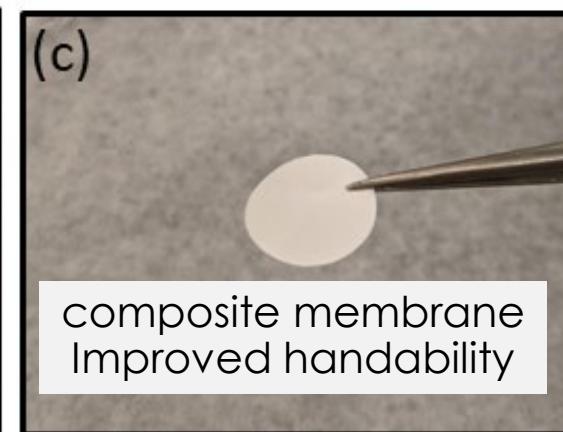


Ceramic = $\text{Li}_{1+x+y}\text{Al}_x\text{Ti}_{2-x}\text{Si}_y\text{P}_{3-y}\text{O}_{12}$
LICGC™ (Ohara) powder

Single-ion gel composite – good t_{Li^+} , stable Li/Li cycles, and handability is good

Transference number

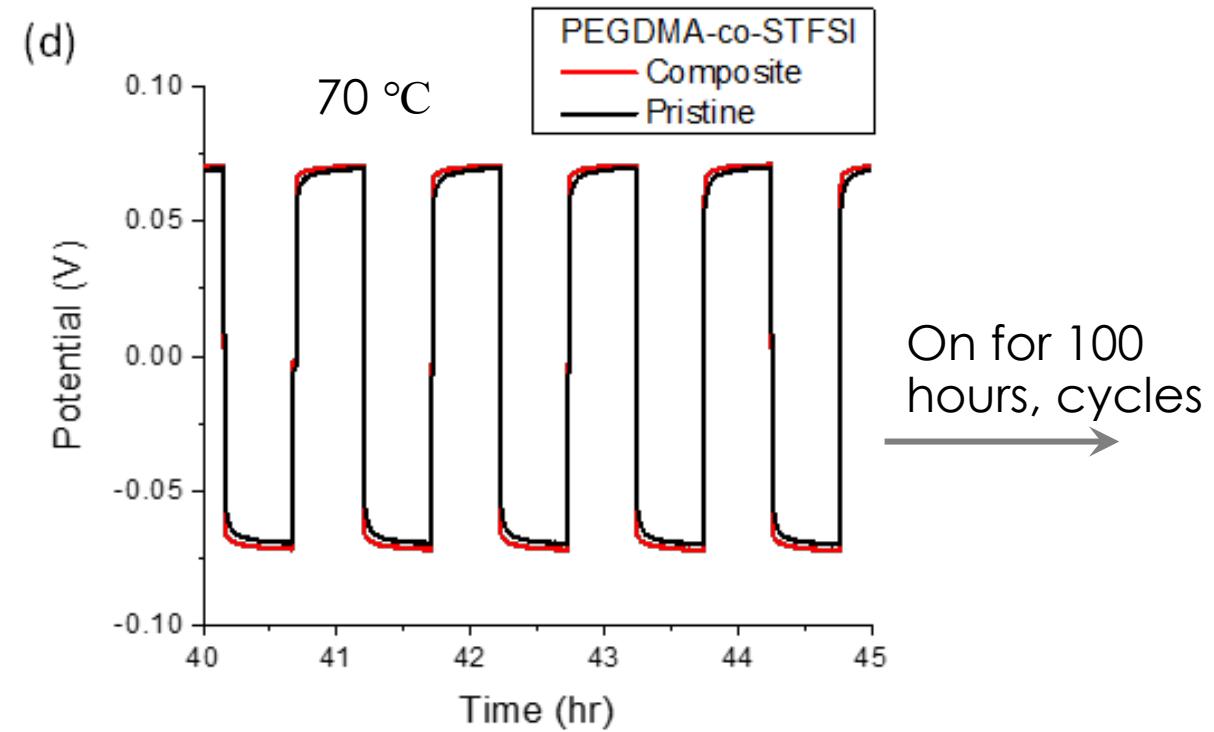
Samples all gel polymers	t_{Li^+}
Single ion polymer – STFSI anion	0.90
Single ion polymer – STFSI anion + 50 wt% LICGC™	0.88
Polymer with free LiTFSI salt	0.28
Polymer with free LiTFSI salt + 50 wt% LICGC™	0.56



Single ion gel polymer vs composite:

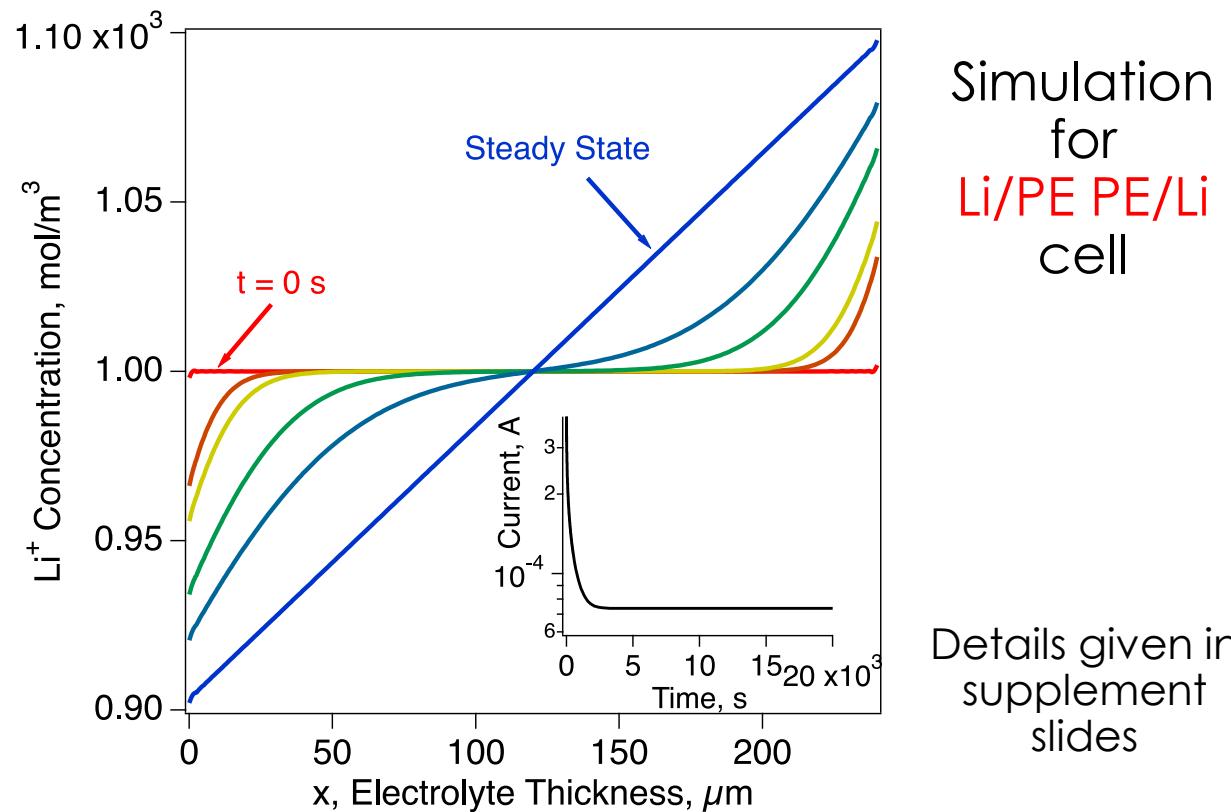
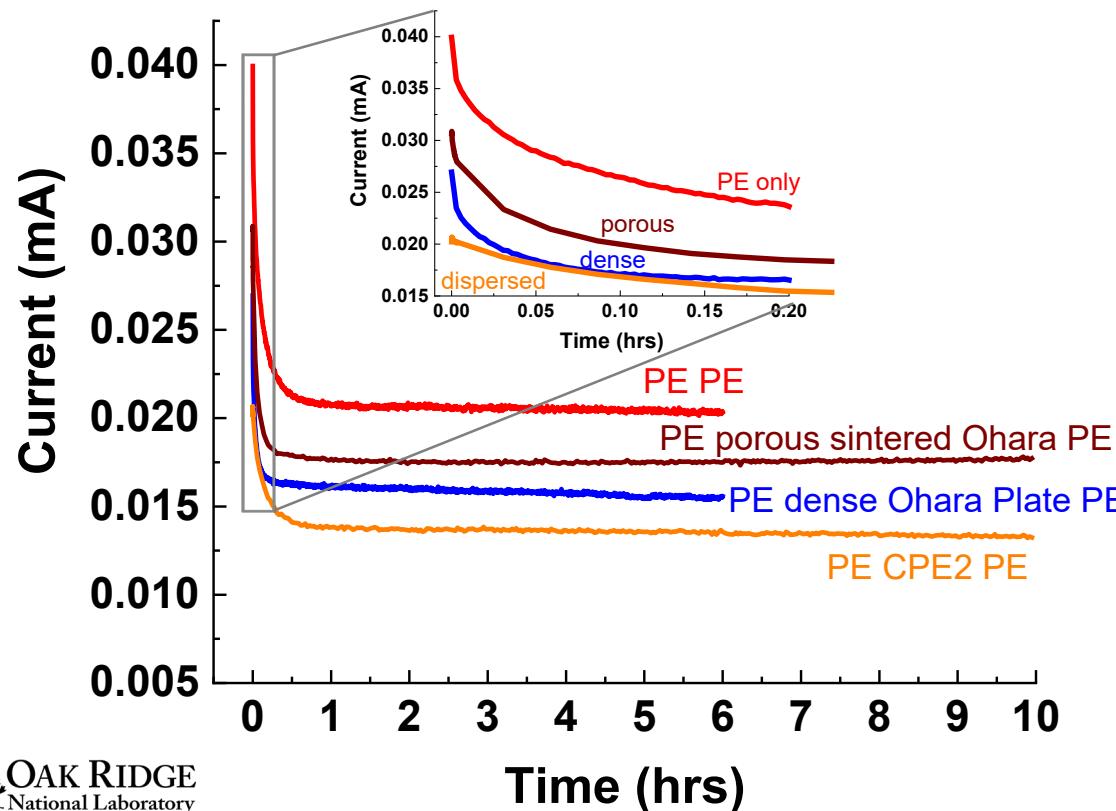
- similar conductivities,
- similar transference numbers
- composite mechanically more robust

Breaking the usual tradeoff



Transference and Li//Li polarization getting a closer look

- Observations of chronoamperometric relaxation with EIS
 - Adding ceramic particles changes the t_{Li^+} for PE (via Bruce Vincent)
 - A SIC blocking layer change kinetics
- Models to assess the relaxation kinetics are being developed

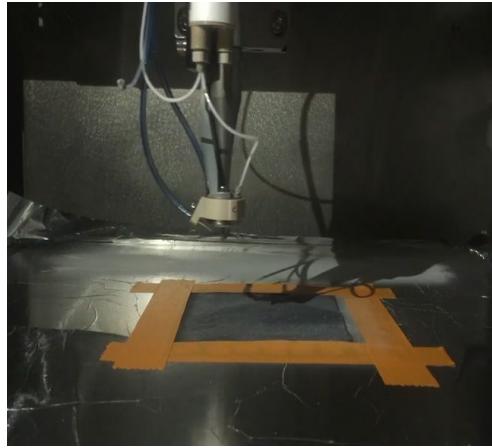


Gel electrolytes with dispersed ceramic particles – perspective, next steps

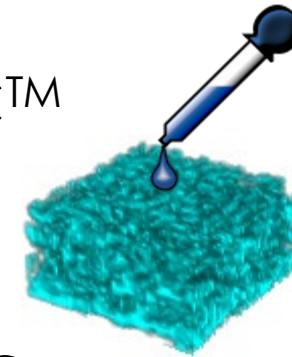
- Benefits: Many ways to tune
 - Single ion conductors may alleviate Li dendrites
- Challenges:
 - Balance of mechanical properties and ion transport, typical tradeoff
 - Cycling tests at practical currents and temperatures
 - Fabrication and integration of thin (~20 μm) membranes
- Future work: Build on what is deduced for ion distribution in CPE:
 - Understand polarization in CPE, tradeoff with conductivity.
 - Assess at microscale, the CPE stability with Li.

Porous sintered ceramic composites

Composites replacing dispersed ceramic particles with sintered particle → best for transport and robust mechanics



1. Spray coat LICGC™ powder suspension.



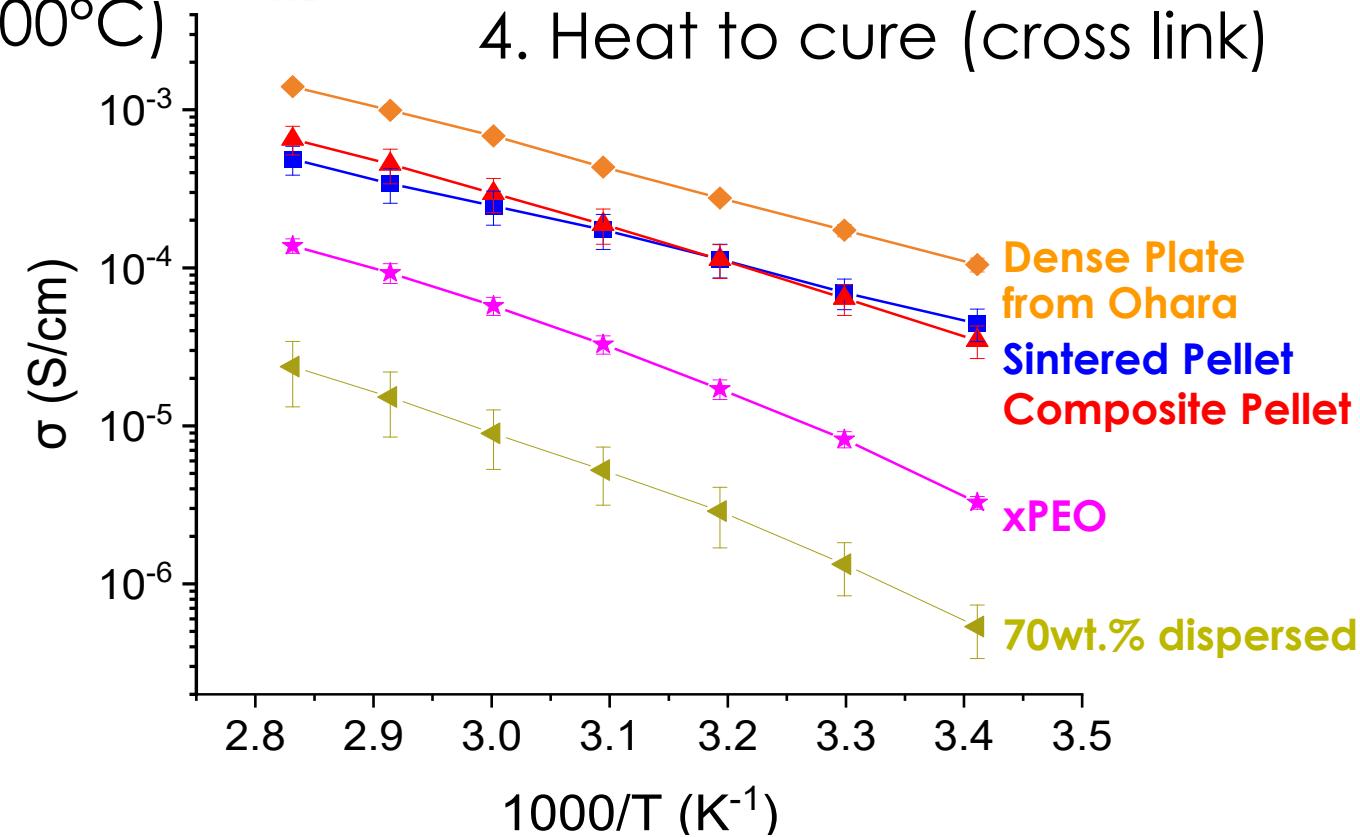
2. Sinter at high temperature (1000°C)

3. Add cross-linkable polymer (Jeffamine + PEGDGE) + LiTFSI under vacuum

4. Heat to cure (cross link)

Properties

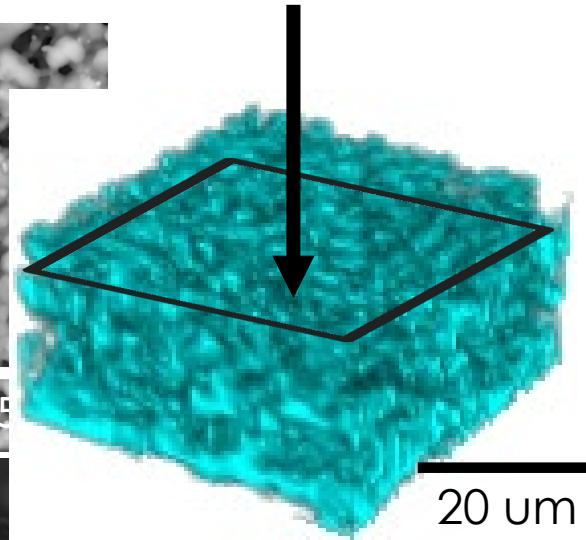
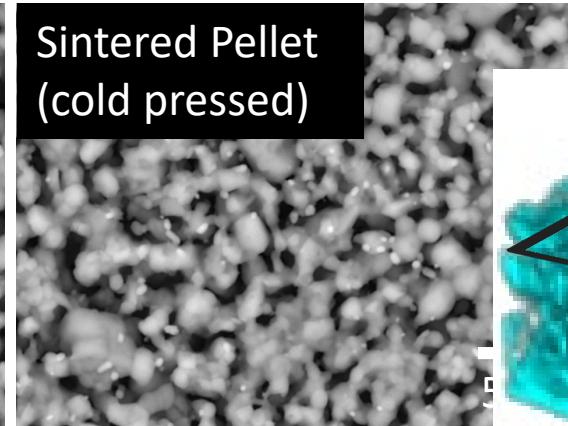
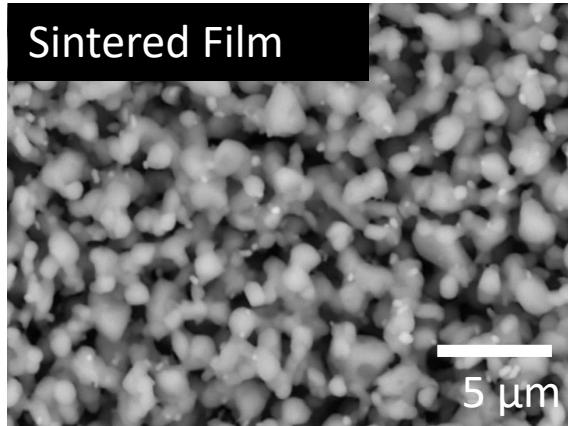
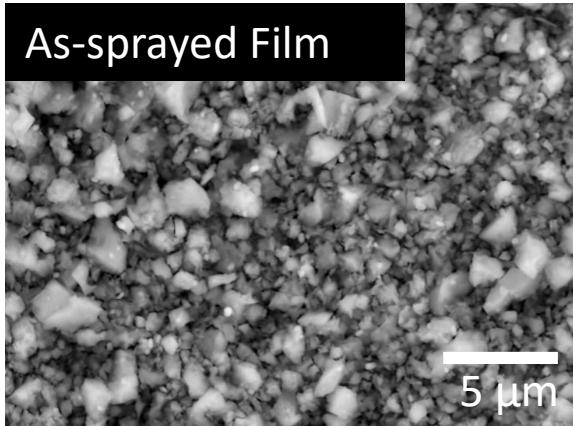
- 77wt.%, 61vol% ceramic
- Conductivity 10^{-4} S/cm
 - Li⁺ path through ceramic phase
 - Grain boundaries are not barriers
- Dry xPEO+LiTFSI phase is resistive
 - Polymer is chemical & mechanical protection



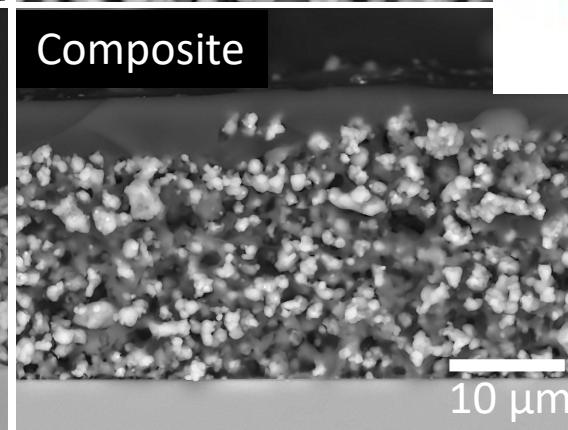
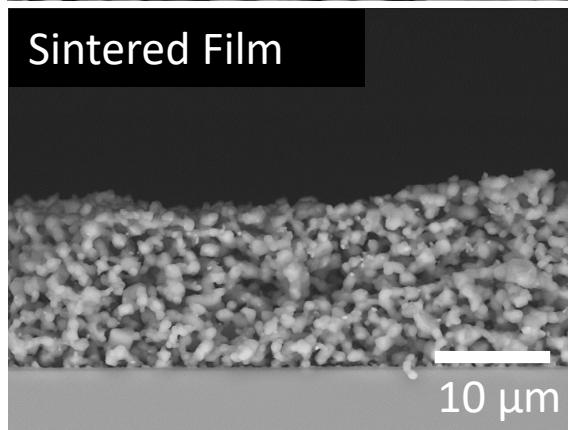
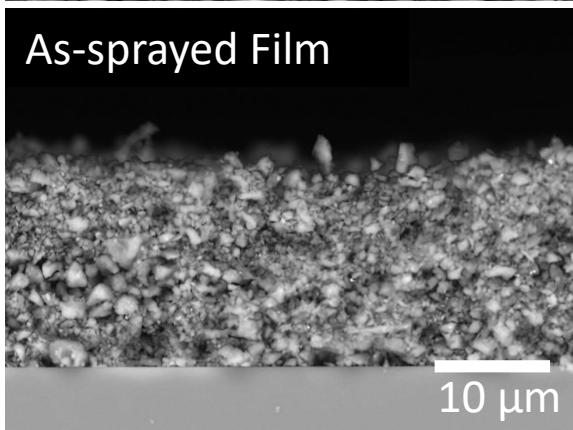
Sintered structure of the ceramic film and composite

Accomplishment

Top view



Edge view

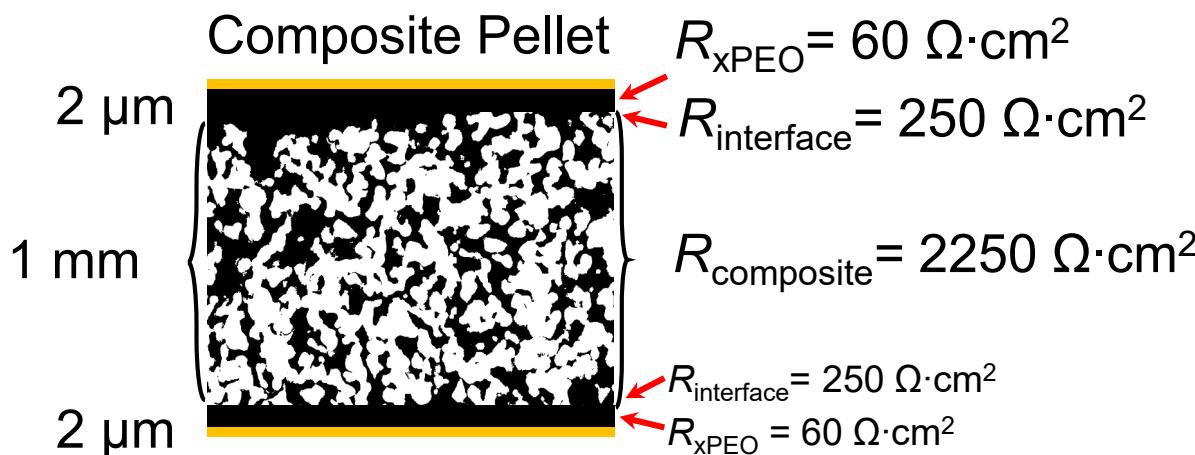
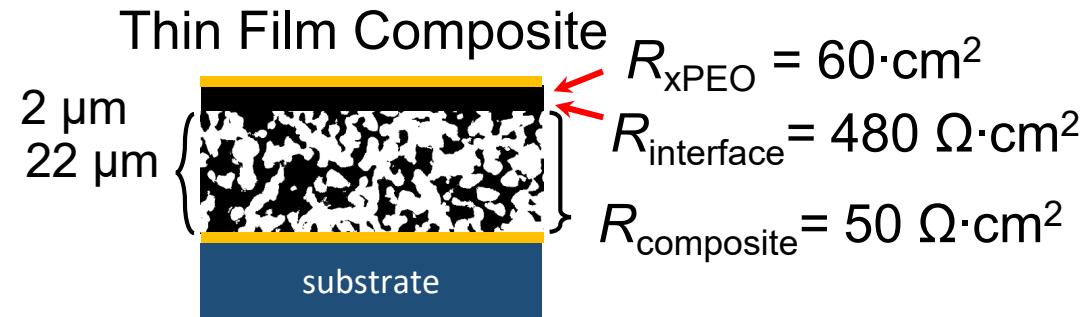


- Sprayed ceramic → uniform close packed particles
- X-ray tomography → homogeneous, partially sintered, fully connected
- Polymer overfill → adds buffer layer for Li contact, but also resistance

Sintered CPE pellets tested for Li cycles and mechanics

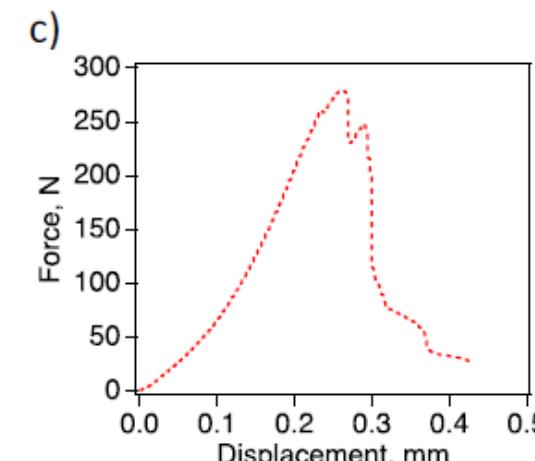
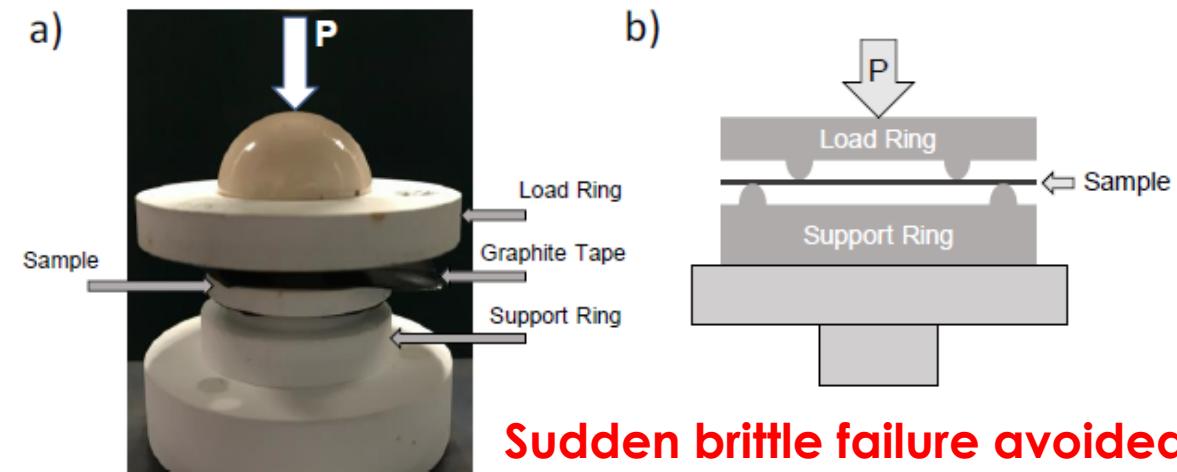
Polymer overlayer

- Needed for $\text{Li}_{1+x+y}\text{Al}_x\text{Ti}_{2-x}\text{Si}_y\text{P}_{3-y}\text{O}_{12}$
- Dominates resistance of thin film



Equi-biaxial flex strength test

- Crack initiates at center of pellet
- Polymer bears some of load

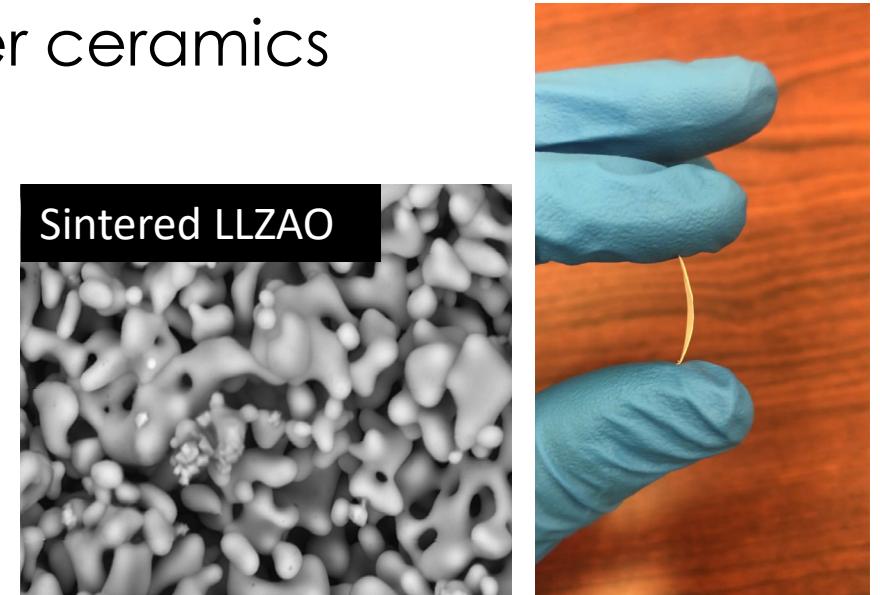


Sudden brittle failure avoided.



Porous sintered composite electrolytes – perspective, next steps

- Benefit: high room-temperature conductivity, mechanically robust
- Challenge: high temperature required to sinter ceramics
- Future work: Focus on thin membranes
 - Alternate materials, processing
 - Test without polymer interface
 - Li/Li symmetrical cell testing
 - Co-sinter with cathode



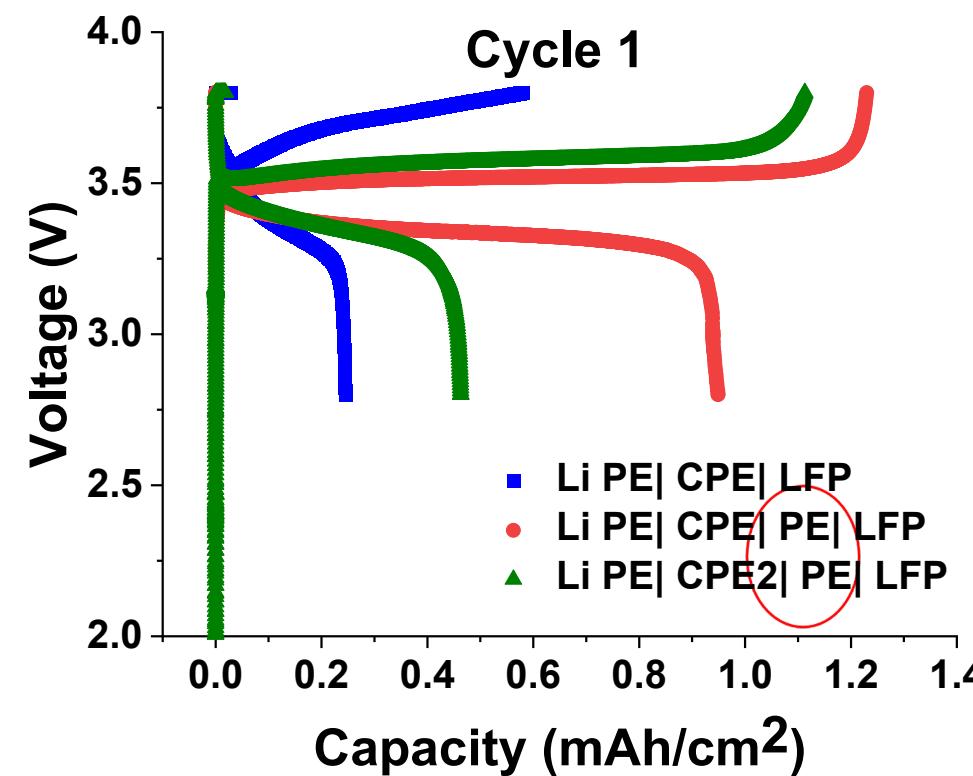
Fabricating and cycling full cells

LiFePO₄ on Cu + **Solid Electrolyte** + **Li metal**
dry composite cathode layers of composites, polymers, ceramics Thick foil or thin film from vapor

Soft polymer is “stopgap” filler between composites with high ceramic loading (LFP and CPE) and at Li interface

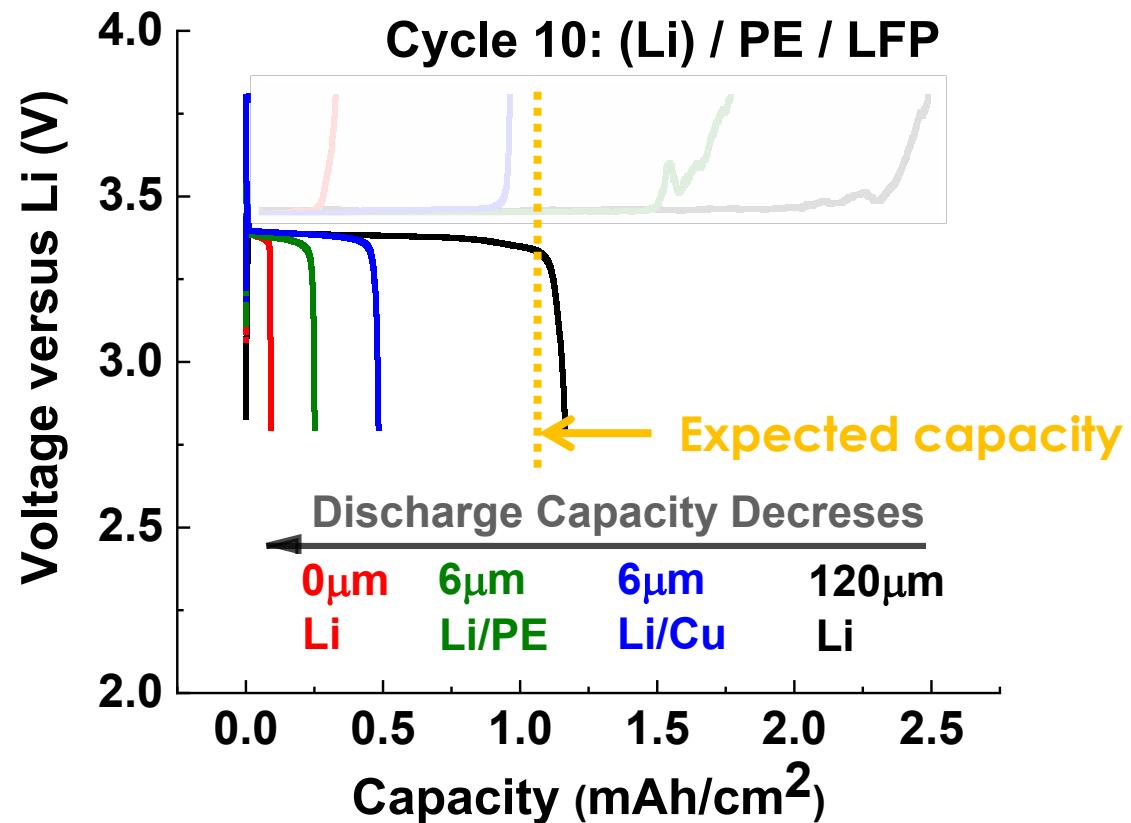
Thin polymer electrolyte (PE) fills gaps at rough LFP / CPE interface

- Improves contact, but adds cell resistance



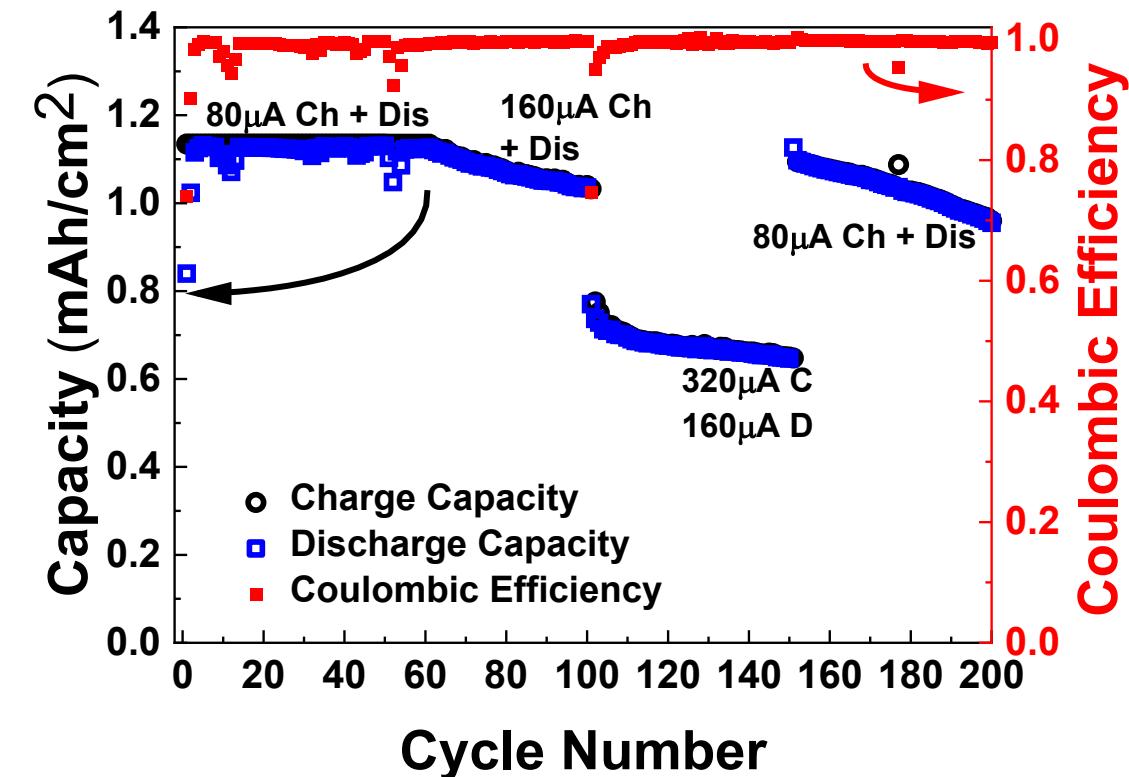
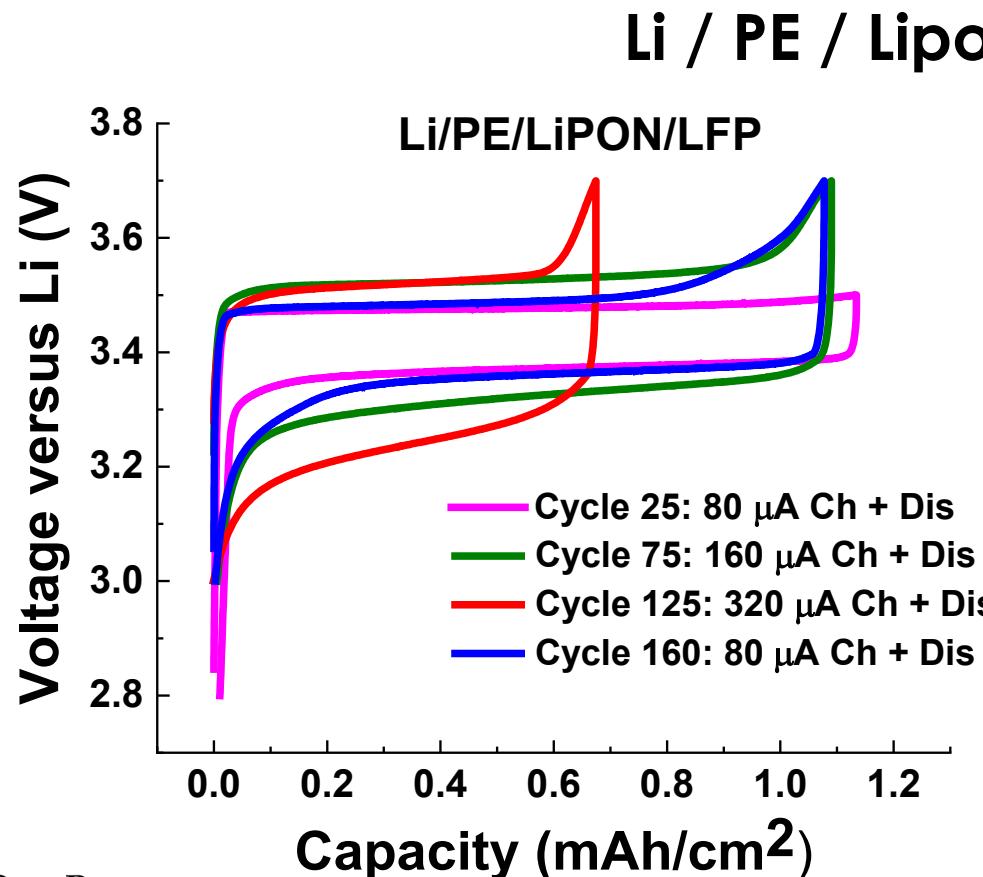
Reducing excess Li, clearly shows if capacity is fading. Here at cycle 10.

- Is Li loss due to physical isolation ? OR
- Is Li loss due to chemical reaction ?



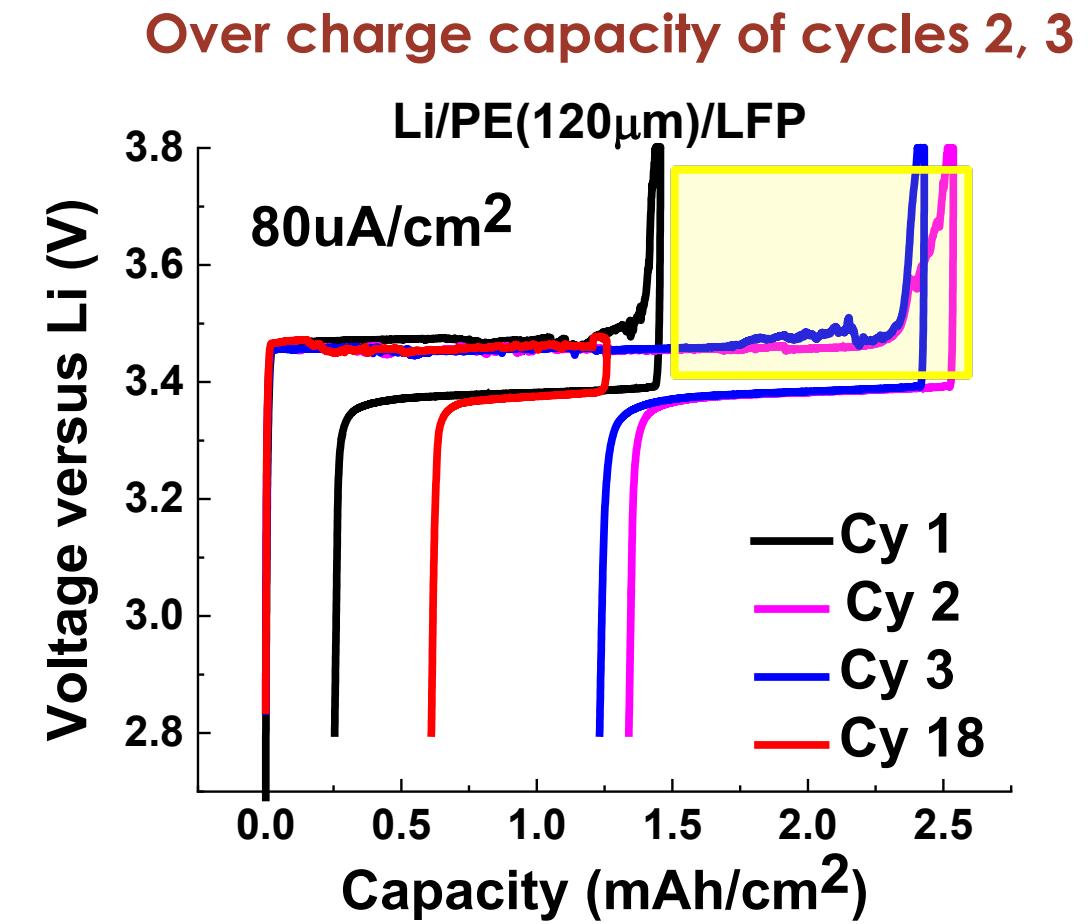
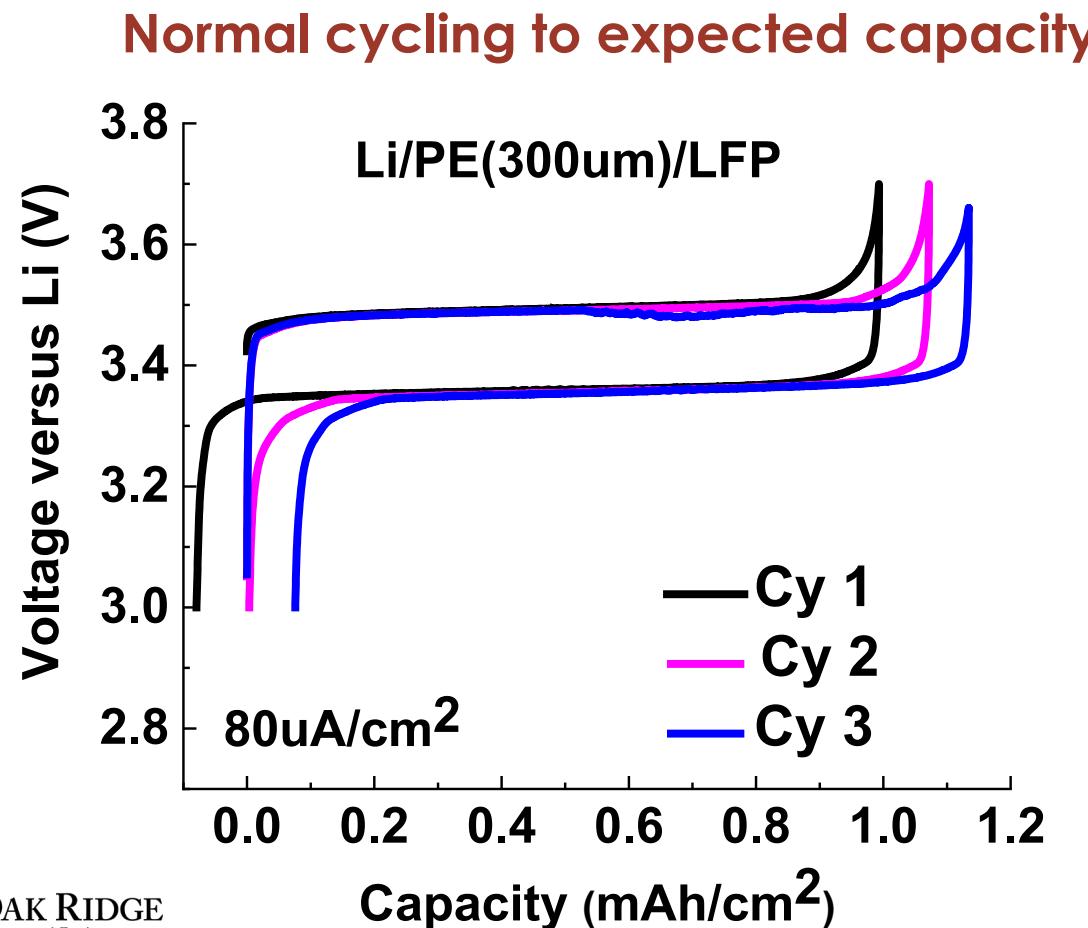
Single ion conduction layer (Lipon) → best cycling to date

- Motivated by work of J. Christensen and N. Craig (Bosch)
- Here stable (although resistive) cell with Lipon sputter-coated cathode
- Long cycling. Also seen for very thick PE ($\sim 300\mu\text{m}$) and thick Li



Overcharging is observed – assessing mechanism(s)

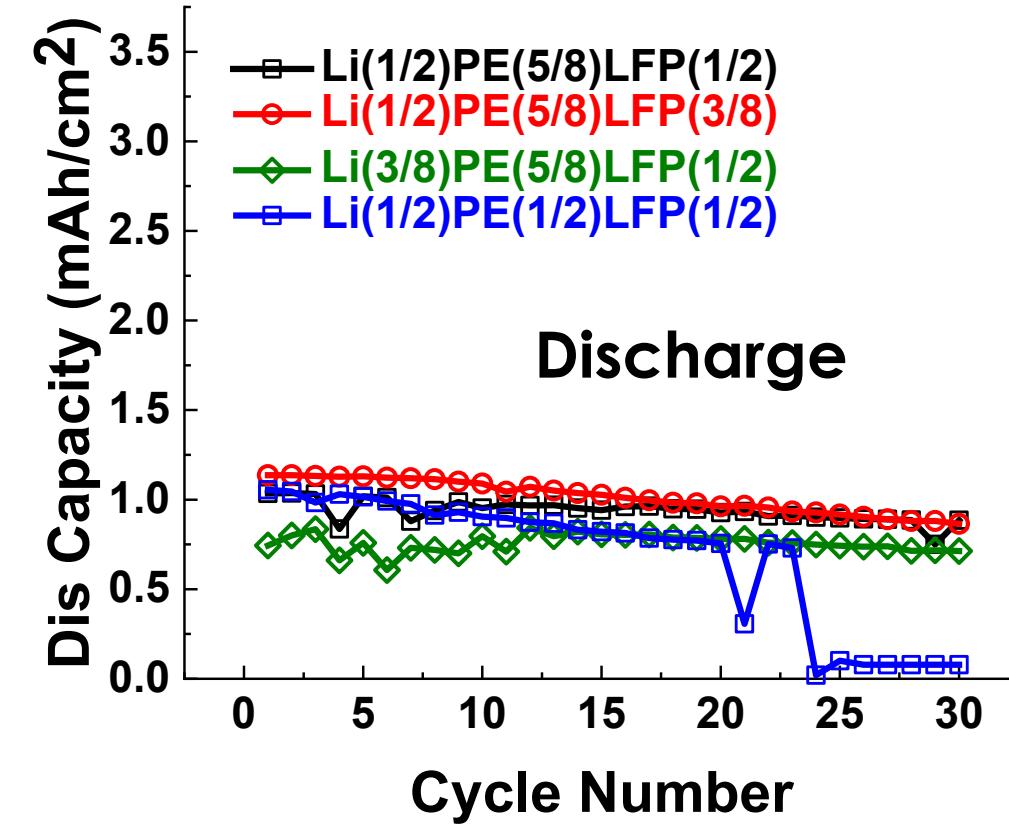
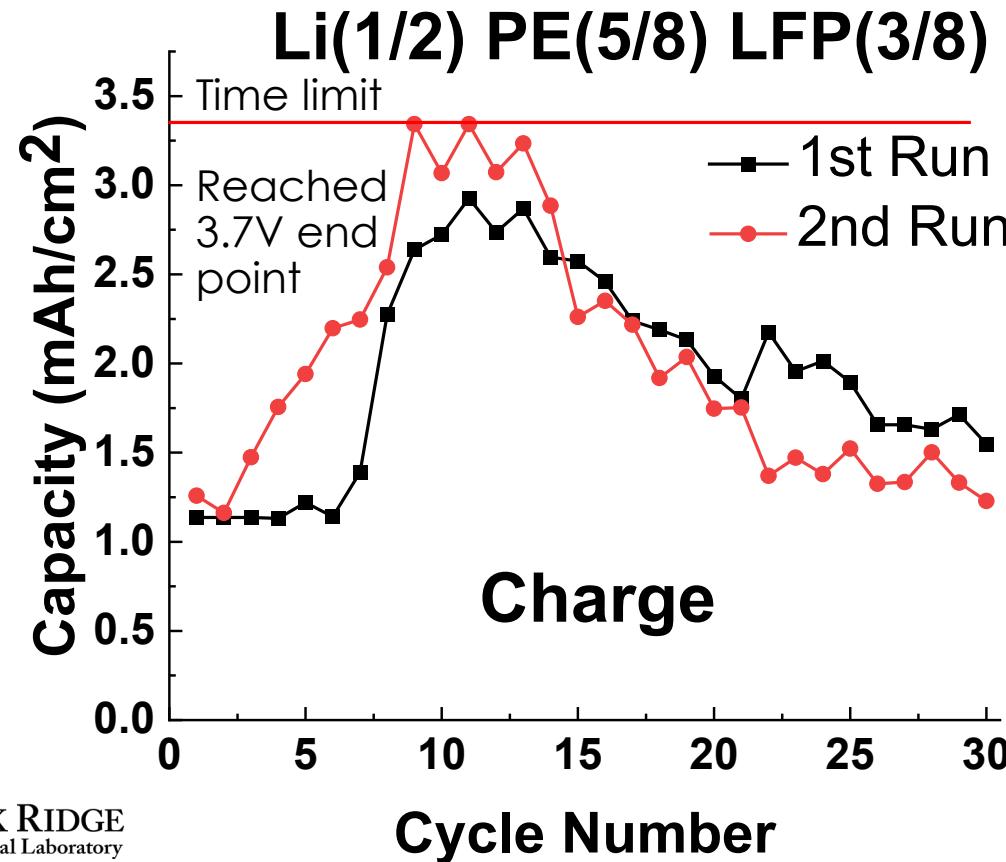
- Over-charge can exceed LiFePO₄ capacity (1.1mAh/cm²) by 2-3X.
- Could this be soft lithium shorts?
- Depends on cell layouts and PE thickness, reproducibly.



Overcharge related to cell design – dimensions, thickness

- Observations in coin cells

- Overcharge related to electrode areas → suggests edges are more sensitive
- Discharge cycles are well behaved → no major damage
- Similar to Homann et al. (Sci. Rept. 2020) → attributed noise to Li micro-dendrite



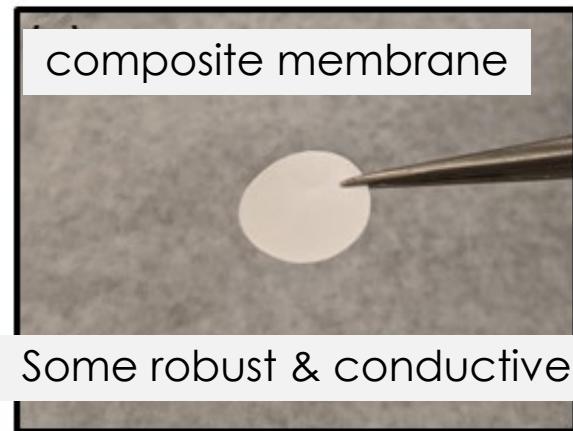
Full cell fabrication and cycling – perspective, next steps

- Benefit: Initial work with LiFePO₄
 - allowed isolation of Li aging, and focus on electrolyte research
- Challenge
 - Identify processes leading to Li loss, degradation
 - Test cycling with practical thicknesses, temperatures, larger layouts
- Future work
 - Test with HE/HV NMC 622 cathodes.
 - Homann et.al. (2020) – PEO based PE should be stable to 4.6V
 - Assess advantages of incorporating SIC electrolyte, best location
 - Alternate processing to form interfaces, eliminate soft-fill layers

Summary- CPE of new materials/processing → path to SSB

Summary with future work

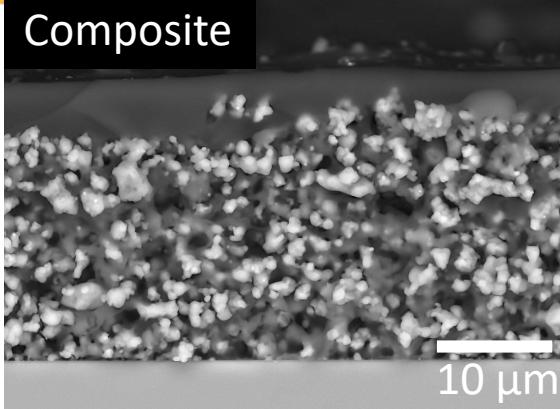
SIC or crosslinked gel with high ceramic loading



Future work

- Tune materials & interfaces ~ free Li⁺
- Polarization in composite
- Reduce thickness

Highly conductive sintered thin-film CPE

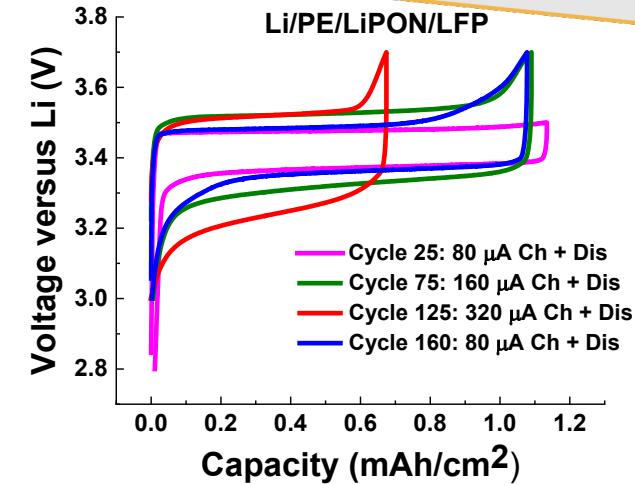


Future work

- Microstructure ~ Li dendrites
- Low temp sintering of thin ceramic
- Integrate with cathode

Focus on high energy density. Thin Li and electrolytes with good interfaces.

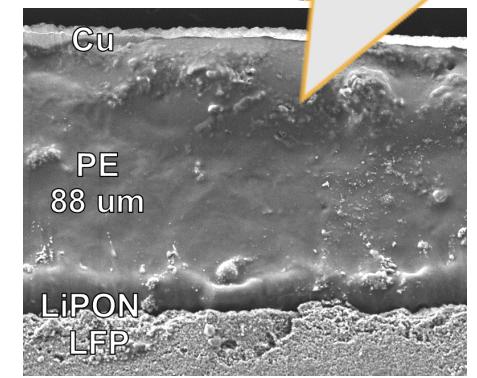
Integration into cell balancing ASR and stability



Future work

- Aging processes at Li interface
- Effective SIC use
- Reduce SE thicknesses
- HE cathode evaluation

Li-free anode to achieve max energy density



Future work

- Processing to form/integrate ultra thin Li

Technical backup slides

- Details of polarization model

Setup of model (for review slides only is needed)

- In order to explicitly include the double layer at the electrode surface we solve Poisson's equation for potential change due to distribution of charge carriers in electrolyte

$$-\epsilon \nabla^2 \phi = F(c_+ - c_-)$$

- Fluxes of charge carriers in the electrolyte are modeled using Nernst-Plank equations

$$N_+ = -z_+ u_+ F c_+ \nabla \phi - D_+ \nabla c_+$$

$$N_- = -z_- u_- F c_- \nabla \phi - D_- \nabla c_-$$

- Mixed boundary conditions need to be prescribed at the electrode surfaces ($z = 0$ and $z = L$)

$$\phi(0) - \lambda_s \nabla \phi(0) = 0$$

$$\phi(L) + \lambda_s \nabla \phi(L) = V,$$

- Evolution of concentration of charge carriers in time (mass balance)

$$\frac{\partial c_+}{\partial t} = -\nabla N_+$$

$$\frac{\partial c_-}{\partial t} = -\nabla N_-$$